Pentadiamond: A Hard Carbon Allotrope of a Pentagonal Network of sp² and sp³ C Atoms

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A pentagonal covalent network consisting of sp² and sp³ C atoms has been investigated based on the density functional theory. Our theoretical investigations clarified that the pentagonal covalent network is a metastable three-dimensional carbon allotrope with the Fm3m space group possessing remarkable mechanical properties: relatively high bulk modulus of 381 GPa together with a negative Poisson’s ratio of ~0.241. Accordingly, the pentagonal covalent network possesses extremely high Young’s and shear moduli of 1691 and 1113 GPa, respectively, surpassing those of the diamond. The electronic structure of the pentagonal network is a semiconductor with an indirect band gap of 2.52 eV between L and X points for valence and conduction band edges, respectively, with the relatively small carrier masses.

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Carbon is a unique element in terms of the structural variation in its possible allotropes [1], owing to the three possible orbital hybridizations, boundary conditions, and topological defects. Three forms of orbital hybridization, sp, sp², and sp³, render allotropes covering all dimensions: C with the sp, sp², and sp³ hybridization forms one-dimensional C chains (polyene) [2,3], two-dimensional sheet (graphene) [4], and three-dimensional network (diamond), respectively, which are well-known conventional carbon allotropes. In addition, the boundary conditions and topological defects cause further allotropes with unique morphologies, such as fullerene [5] and carbon nanotubes [6], which possess unusual electronic and structural properties depending on both local and global network topologies [7–14]. For instance, fullerenes basically possess spherical harmonic YLm(θ, φ) electron states with the deep lowest unoccupied states owing to their hollow cage network topology caused by 12 pentagonal rings embedded in appropriate number of hexagons [7–11]. Moreover, networks consisting of pentagonal rings of sp² or sp³ C atoms causes unusual physical properties, such as magnetism on two dimensional sp² C networks [15] and negative Poisson’s ratio on pentagraphene [16,17].

Carbon allotropes consisting of both sp² and sp³ C atoms have been attracting much attention because of their morphological diversity, which is caused by the huge number of combinations and of arrangements of sp² and sp³ C atoms in the networks. High-pressure and high-temperature treated C₆₀ and higher fullerenes are representative examples of such allotropes, being expected to possess low mass density owing to their covalent bonds between fullerenes and nanoscale pore arising from fullerene cages [18–21]. The rapid quenching of amorphous carbon causes a sp² and sp³ hybrid C allotrope, known as the Q carbon, which has been reported to possess magnetism and remarkable hardness surpassing diamond [22,23]. In addition to these all carbon materials, the controlled polymerization of hydrocarbon molecules causes two- and three-dimensional covalent hydrocarbon networks of sp² and sp³ C atoms by assembling appropriate linker and linkage hydrocarbon molecules [24–28]. Such bottom-up synthesis techniques ensure that the novel carbon allotropes with unusual physical properties are expected by assembling and polymerizing appropriate hydrocarbon molecules [29,30]. These morphologically and structurally well-defined covalent hydrocarbon networks are considered to be applicable for wide area of modern technologies, such as gas storage, catalytic application, and optoelectronics, because of their porous structure with stable covalent framework and unusual electronic structures.

In this Letter, we aim to theoretically explore a three-dimensional carbon allotrope consisting of sp² and sp³ C atoms (pentadiamond), owing to copolymerization of hydrocarbon molecules containing pentagonal rings (spiro[4.4]nona-2,7-diene [31,32] and [5.5.5.5]fenestra-trena [33–36]) as for a possible candidate of hard carbon allotropes with unusual mechanical properties, using the density functional theory with the generalized gradient approximation. The pentadiamond sorely consists of pentagonal rings with the Fm3m space group owing to the copolymerization of these constituent molecules. Our calculations demonstrated that the pentadiamond has the high bulk modulus of 381 GPa, which is approximately 80% of that of diamond, indicating that the pentadiamond is the hard carbon allotrope. Furthermore, it has negative Poisson’s ratio of ~0.241, leading to extremely high...
Young’s and shear moduli of 1691 and 1113 GPa, respectively, which is higher than those of diamond and other ultrahard carbons. The pentadiamond is a semiconductor with an indirect band gap of 2.25 eV expected to possess high hole mobility.

All calculations were conducted on the basis of the density functional theory [37,38] implemented in the program package STATE [39–41]. We used the generalized gradient approximation with the Perdew-Burke-Ernzerhof functional to describe the exchange-correlation potential energy among interacting electrons [42]. A ultrasoft pseudopotential generated using the Vanderbilt scheme is adopted for describing the interaction between electrons and ions [43]. Valence wave functions and deficit charge densities were expanded in terms of plane-wave basis sets with cutoff energies of 25 and 225 Ry, respectively, which give sufficient convergence in the total energy and electronic structures of covalent networks consisting of both sp$^2$ and sp$^3$ C atoms [44]. Ab initio molecular dynamics simulations were conducted using the velocity scaling method to keep the temperature constant during the simulation to insure the thermal stability of the pentadiamond. Integration over the first Brillouin zone was carried out using equidistant 4 × 4 × 4 and 8 × 8 × 8 k meshes for the self-consistent electronic structure calculations and for the density of states calculations, respectively. Lattice parameters and internal atomic coordinates were fully optimized until the force acting on each atom became less than 5 mRy/Å.

Figure 1 shows an optimized geometry of the pentadiamond with the lattice parameter of 9.195 Å and the Fm$\overline{3}$m space group. The covalent network only consists of pentagons in which three of five edges are shared by its adjacent pentagons [Fig. 1(a)], owing to the copolymerization of spiro[4.4]nona-2,7-diene and [5.5.5.5]fenestratetraene which are alternately arranged at the vertexes of the cubic lattice [Fig. 1(b)]. According to these constituent molecules and their arrangement, a unit cell of the pentadiamond contains 22 C atoms, 10 and 12 of which are sp$^3$ and sp$^2$ C atoms, respectively. In addition, the symmetry operators belonging to the Fm$\overline{3}$m group reduce the independent atomic sites to three, whose atomic coordinates are listed in Table I. Because the network consists of sp$^2$ (threefold coordinated) and sp$^3$ (fourfold coordinated) C atoms, covalent bonds are classified into two groups. Calculated bond lengths associated with the sp$^3$ C atom are 1.563 and 1.520 Å for C1-C1 and C2-C3 bonds, respectively, reflecting their sp$^3$ bond nature. In contrast, the bond length associated with sp$^2$ C atoms (C3-C3 bond) is 1.349 Å, indicating their double bond nature of sp$^2$ C atoms. As shown in Fig. 1(a), the pentadiamond possess large cubic pores with 3.644 Å edges surrounded by the pentagonal covalent network. Therefore, it has a low mass density with the density of 2.26 g/cm$^3$, which is the same as that of graphite while smaller by 36% than that of diamond.

The pentadiamond has relative total energy of 275 meV/atom with respect to that of diamond (Fig. S1 [45]). The total energy is higher than that of diamond, graphite, and other hard carbon materials (Table II), but still lower than that of C$_{60}$ which is known to be metastable zero-dimensional carbon allotrope. The moderate total energy is ascribed to the structural distortion of bond angles for both sp$^2$ and sp$^3$ C atoms. For the sp$^3$ C atoms, although the C1 atom has almost perfect sp$^3$ hybridization with the bond angle of $\theta_{212} = 109.4^\circ$, the C2 atom has bond angles of $\theta_{123} = 115.9^\circ$ and $\theta_{323} = 101.9^\circ$, which are wider and narrower than the bond angle of an ideal sp$^3$. For the sp$^2$ C atoms, because of the pentagonal network, the bond angle associated with the C3 is $\theta_{232} = 133.4^\circ$ and $\theta_{233} = 113.3^\circ$ which are significantly larger and smaller, respectively, than the sp$^2$ bond angle. The moderate energy also increases the formation energy $\Delta E$ of the pentadiamond on

![FIG. 1](a) An optimized geometry of the pentadiamond under an optimum lattice constant of 9.195 Å and the Fm$\overline{3}$m space group. Pale and dark gray balls indicate sp$^3$ and sp$^2$ carbon atoms, respectively. Indexes C1, C2, and C3 indicate the independent atomic site under the Fm$\overline{3}$m space group. (b) Spiro[4.4]nona-2,7-diene and [5.5.5.5]fenestratetraene are alternately arranged at the vertexes of the cubic lattice. White and gray balls indicate H and C atoms, respectively.

| TABLE I. Fractional atomic coordinates of the pentadiamond with the Fm$\overline{3}$m space group under the optimized lattice parameter of 9.195 Å. Atomic indexes are defined in Fig. 1(a). |
|-----------------|-----------------|-----------------|
| C1 | 0.250 | 0.250 | 0.250 |
| C2 | 0.152 | 0.152 | 0.152 |
| C3 | 0.198 | 0.198 | 0.000 |
the direct copolymerization of spiro[4.4]nona-2,7-diene and [5.5.5.5]fenestratetraene:

\[
\text{spirononadiene} + \text{fenestratetraene} = \text{pentadiamond} + 12\text{H}_2 + \Delta E
\]

The calculated formation energy is 0.31 eV/atom, reflecting the energy cost to form covalent network of sp\(^2\) and sp\(^3\) C with distorted bond angles. Thus, the pentadiamond is expected to be synthesized using the Yamamoto-type Ullmann cross coupling [26] on bromospiro[4.4]nona-2,7-diene and bromo[5.5.5.5]fenestratetraene instead of their pristine form as the cases of polymerization reactions of hydrocarbons. Furthermore, the self-organization process of the precursors with appropriate functional groups in liquid is the other possible procedure as the case of covalent organic frameworks [24].

We investigate the thermal stability of the pentadiamond by conducting \textit{ab initio} molecular dynamics (MD) simulations under the temperature of 4000 K. To explore the possibility of long-range structural distortions, we perform the constant-temperature MD calculations on the extended cell, which contains 88 C atoms, and the primitive \(1 \times 1 \times 1\) cell for 14 and 146 psec simulation times, respectively. After the 14 psec MD simulation on the pentadiamond with the extended cell at the temperature of 4000 K, the pentadiamond keeps its initial covalent network topologies. As for the primitive cell simulation, the pentadiamond also retain their covalent network up to 146 psec simulation time. In addition, there are no soft vibration modes in the phonon spectra, indicating that the pentadiamond is dynamically stable (Fig. 2). The phonon dispersion calculation was performed with density functional perturbation theory [53] implemented in the QUANTUM EXPRESSO code [54]. Therefore, the pentadiamond is thermally and energetically stable, once it was synthesized using appropriate experimental schemes.

The pentadiamond possesses remarkable mechanical properties summarized in Table II. The mechanical properties are studied using the elastic constants (stiffness) \(c_{ij}\), which is evaluated by taking the finite difference of the total energy with respect to strains. Calculated elastic constants are 1715.3, \(-283.5\), and 1187.5 GPa for \(c_{11}(=c_{22}=c_{33})\), \(c_{12}(=c_{13}=c_{23})\), and \(c_{44}(=c_{55}=c_{66})\), respectively. Note that the elastic constants obviously satisfy the Born stability criterion \(c_{11} - c_{12} > 0\), \(c_{11} + 2c_{12} > 0\), and \(c_{44} > 0\), corroborating the structural stability of the pentadiamond. Under the cubic symmetry, the bulk modulus \(B\) is calculated using the form

\[
B = \frac{c_{11} + 2c_{12}}{3}.
\]

The calculated bulk modulus \(B\) is 381 GPa which is over 80% of that of diamond and close to those of the other hard and ultrahard carbon materials (Table II), indicating the pentadiamond is a potential candidate of hard carbon allotropes, even though it has low density as that of graphite.

![FIG. 2. Phonon dispersion and phonon density of states of pentadiamond.](image-url)
To further investigate the mechanical properties, we calculate the Young’s modulus of the pentadiamond using the equation,

\[
E = \frac{1}{s_{11} - 2(s_{11} - s_{12} - \frac{s_{44}}{2}) (\cos^2 \theta \sin^2 \phi + \sin^2 \theta \cos^2 \phi \sin^2 \phi)},
\]

where \(\theta\) and \(\phi\) are the Euler’s angles and \(s_{ij}\) are the elastic compliance evaluated by \(c_{ij}\) with relations of \(s_{11} = [c_{11} + c_{12}]/[(c_{11} - c_{12})(c_{11} + 2c_{12})], s_{12} = [-c_{12}]/[(c_{11} - c_{12})(c_{11} + 2c_{12})]\), and \(s_{44} = 1/c_{44}\). The calculated Young’s modulus is shown in Fig. 3(a), indicating that the pentadiamond has extremely high Young’s modulus over 15 TPa for all directions. Using the Young’s and bulk moduli, we further evaluated the shear modulus [Fig. 3(b)]. As the Young’s modulus, the pentadiamond also possesses the high shear modulus over 1 TPa for all directions. Therefore, the pentadiamond may exhibit extreme stiffness against anisotropic structural deformations. Note that the Young’s and shear moduli are higher than those of other hard and ultrahard carbon allotropes (Table II). These large moduli imply that the pentadiamond possesses the negative Poisson’s ratio. Indeed, the calculated Poisson’s ratio exhibits negative values ranging from \(-0.20\) to \(-0.28\), depending on the lattice directions [Fig. 3(c)]. Furthermore, the remarkable mechanical properties causes the high sound speed of 28700 m/s, which is higher than those of the diamond and other bulk carbon materials. Because the sound speed is expressed as \(v_s = \sqrt{(3B/\rho)[(1-\nu)/(1+\nu)]}\) where \(B\), \(\rho\), and \(\nu\) are the Bulk modulus, density, and Poisson’s ratio, respectively, the remarkable sound speed is ascribed to the negative Poisson’s ratio. Accordingly, the pentadiamond is a hard carbon allotrope with unusual mechanical response on the uniaxial and shear strain, being expected to constituent material for wide area of modern technologies.

Figure 4 shows the electronic structure and density of states of the pentadiamond. The pentadiamond is a semiconductor with an indirect band gap of 2.52 eV. The valence band top and the conduction band bottom are located at the \(L\) and \(X\) points, respectively. The highest branch of the valence band and the lowest branch of the conduction band possess the substantial dispersion of 1 eV or more. Therefore, the pentadiamond is expected to have small effective masses around its band edges: calculated electron masses at the \(X\) are 0.98 and 0.67 \(m_e\) along directions to the \(\Gamma\) and \(W\) points, respectively, which are lighter than that of the diamond. In contrast, as for the valence band edge, calculated hole masses at the \(L\) point are 1.59 and 0.76 \(m_e\) along directions to the \(W\) and \(\Gamma\) points, respectively, being heavier than that of the diamond. The moderate carrier mass and the large density of states at the band edges imply that the pentadiamond may possess moderate carrier mobility for both electron and hole. The dispersion relation as well as the density of state show that the pentadiamond possesses three-dimensional bulk electronic structure as that of diamond, reflecting its three-dimensional covalent network with the high symmetry.

To provide further insight into the electronic structure of the pentadiamond, we investigate the squared wave function of the highest branch and the lowest branch of the
pentadamond at particular symmetry points (Fig. 5). The wave functions of the highest branch of the valence band at the \( L \) and \( \Gamma \) points are distributed on the C3 atoms with bonding \( \pi \) state nature, owing to their sp\(^3\) hybridization. The wave function of the lowest branch of the conduction band at the \( \Gamma \) and \( X \) points is also distributed on C3 atoms with antibonding \( \pi \) nature. Note that both the valence and conduction states are not pure \( \pi \) state but hybridized states containing small amount of \( \sigma \) component. These facts imply that the electron states near and around the band edges are regarded as the \( \pi \) electron states of sp\(^2\) C dimers which are separated by about 2.6 Å from its eight adjacents. The large band dispersion is ascribed to the substantial overlap of \( \pi \) states between adjacent dimers of C3 atoms thanks to the relatively small spacing between them compared with that of the conventional molecular solids with \( \pi \) electrons.

We theoretically explore the possibility of the three-dimensional allotrope of sp\(^2\) and sp\(^3\) C atoms by assembling adequate hydrocarbon molecules as the constituents, using density functional theory with generalized gradient approximation. We found that the copolymerization of spirom[4.4]nona-2,7-diene and [5.5.5.5]fenestra[5.5.5]hendron causes the three-dimensional covalent pentagonal network of sp\(^2\) and sp\(^3\) C atoms named the pentadamond as the metastable carbon allotrope. The pentadamond has the moderate total energy of 275 meV with respect to that of diamond and remarkable thermal stability up to 4000 K. According to covalent network of pentagonal ring, the pentadamond are possible candidate of the hard carbon material with the remarkable mechanical properties: the bulk modulus is close to that of the diamond, and the Young’s and shear moduli are higher than these of diamond, although it has remarkably low density which is the same as graphite. Furthermore, the pentadamond has negative Poisson’s ratio of −0.24. The pentadamond is a semiconductor with an indirect band gap of 2.52 eV between the \( L \) and \( X \) points for the valence and conduction band edges, respectively, where the moderate carrier mobility is expected by the carrier effective masses and the density of states at the band edges. The detailed wave function analysis unraveled that the electron states near the band edges are ascribed to the \( \pi \) electrons distributed on sp\(^2\) C atoms.

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